Study on NO selective adsorption performance of Rh/MOF-177

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Abstract

In this paper, the NO adsorption performance of Rh/MOF-177 was studied by pulse method. Based on further confirming the literature molecular simulation results which NO, O_2 and CO_2 were adsorbed very little and N_2 was hardly adsorbed, it was found that the MOF-177 material showed great carrier function, and promoted the atomic-sized high dispersion and stability of almost supported Rh particles on MOF-177. The adsorption properties of NO was greatly improved after Rh was introduced into MOF-177 which the adsorption capacity of Rh/MOF-177 with 3% Rh loading can reach about $310 \times 10^{-3} \text{ mmol}[?]g^{-1}$ by pulse method and was about 15-20 times higher than that of O_2 and CO_2 with near 90% adsorption selectivity in simulated flue gas atmosphere. The NO adsorption capacity of Rh/MOF-177 was related to the reduced Rh⁰ which one reduced Rh⁰ adsorbed three NO molecules, and NO adsorption form on Rh/MOF-177 should be a trinitro-group (Rh(NO)₃) species.

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Highlights

- Almost atomic-sized Rh particles with high dispersion and high stability were firstly prepared by loading method on MOF-177, the Rh/MOF-177 material showed high NO adsorption capacity and adsorption selectivity, and preliminarily exhibited the potential of adsorptive-separating purification and resource utilization of atmospheric pollutant NO in flue gas.
- The experiment verified the literature molecular simulation results of MOF-177 that NO, O₂ and CO₂ were adsorbed very little and N₂ was hardly adsorbed.
- The structure-function relationship between the chemical and dispersion state of Rh/MOF-177 and NO adsorptive properties was obtained. The NO adsorption capacity of Rh/MOF-177 was related to the reduced Rh⁰ and increased proportionally with the increase of Rh loadings. One reduced Rh⁰ adsorbed three NO molecules and the adsorption form of NO on Rh/MOF-177 should be the adsorbed trinitro group species Rh(NO)₃.

Abstract: In this paper, the NO adsorption performance of Rh/MOF-177 was studied by pulse method. Based on further confirming the literature molecular simulation results which NO, O_2 and CO_2 were adsorbed very little and N_2 was hardly adsorbed, it was found that the MOF-177 material showed great carrier function, and promoted the atomic-sized high dispersion and stability of almost supported Rh particles on MOF-177. The adsorption properties of NO was greatly improved after Rh was introduced into MOF-177 which the adsorption capacity of Rh/MOF-177 with 3% Rh loading can reach about $310 \times 10^{-3} \text{ mmol}[?]g^{-1}$ by pulse method and was about 15-20 times higher than that of O_2 and CO_2 with near 90% adsorption selectivity in simulated flue gas atmosphere. The NO adsorption capacity of Rh/MOF-177 was related to the reduced Rh^0 which one reduced Rh^0 adsorbed three NO molecules, and NO adsorption form on Rh/MOF-177 should be a trinitro-group (Rh(NO)₃) species.

Keywords: MOF-177;Rh component;NO adsorption properties;dispersed state;Structure-function relationship

0 Introduction

At present, nitrogen oxide (NO_x) pollutants exhausted by flue gas of power plants and industrial kilns are mainly treated by selective catalytic reduction technology with an usual supported catalysts loaded on titanium dioxide and molecular sieve as carriers, which need to consume a large amount of reducing agent NH₃ or urea, and the engineering operation cost was generally higher ^[1-2]. In recent years, with the development of new metal organic framework porous materials, the adsorption and separation of NO_x in the flue gas has aroused the interest of researchers, intending to adsorb, separate and enrich the NO_x component in the flue gas and then produce nitrate compounds such as fertilizer by desorption, oxidation and absorption, so as to achieve the resource utilization of nitrogen oxide (NO_x) pollutants. For example, Yang et al.^[3] used MFM-520 metal organic framework materials (metal nodes were divalent Zn ions, organic ligands were 4,4-bipyridyl-3,3',5,5'-tetracarboxylate) to achieved efficient adsorption of NO₂. At 298 K and 0.001MPa, the adsorption amount of NO₂ reached 4.2 mmol*g⁻¹, and the IAST theoretical selectivity to CO₂ was as high as 675, at the same time, the adsorbed NO₂ could be converted into HNO₃ quantitatively.

More than 90% of the nitrogen oxides (NO_x) in the flue gas exist in the form of NO component ^[1,4]. It has a great engineering significance to study the adsorption and separation of NO. Therefore, the adsorption and storage and separation of NO gas by MOFs has been extensively studied^[5-11]. Some MOFs materials show good NO adsorbing effect through the opening metal sites and amine group functionalization. For example, the unsaturated copper atomic metal sites of HKUST-1 store and release medical NO, and the adsorption capacity of NO at room temperature and pressure was 3 mmol*g⁻¹^[6]. The NO adsorption capacity of Ni₂(dhtp)(H₂O)₂*8H₂O (dhtp: 2, 5-dihydroxyterephthalic acid) MOFs material was 6.1 mmol*g^{-1 [10]}. In general, the NO adsorption capacity of Co-CPO-27 and Ni-CPO-27 MOFs was about 6~7 mmol*g⁻¹ [7-8, 12-13]. After amine functionalization of MOFs materials, there was an adsorption chemical reaction between it and NO to form N-diazeniumdiate (NONOate) structure, and then conducted NO adsorption at metal sites. Therefore, MOFs also has a relatively large NO adsorption capacity, such as IRMOF-3 and UMCM-1-NH₂NO, which respectively have 6.4 mmol^{*}g⁻¹ and 1.67 mmol^{*}g⁻¹^[14]. The co-adsorption characteristics under mixed atmosphere by density functional theory (DFT) calculations showed that the NO adsorption capacity of Cu-BTC reached the maximum (10 mmol*g⁻¹) when the adsorption pressure was 5 atm at room temperature, but the CO_2 adsorption capacity also reached the maximum (21 mmol*g⁻¹) at 50 atm adsorption pressure, in other words, compared with NO, more CO_2 was adsorbed by Cu-BTC with the adsorption pressure increasing, and the adsorption selectivity of NO became poor $^{[15]}$. CO₂ adsorbed on M-MOF-74 (M = Mg, Co, Ni) was difficult to be replaced by NO molecules, although NO molecules had a higher binding energy than CO_2 at the MOF-74 metal sites^[16]. In general, MOFs materials have shown excellent NO_x adsorption and storage performance at the beginning. However, current studies mainly focus on the storage and adsorption of NO gas molecules with high purity, such as biomedical NO gas, which is at a disadvantage in competitive adsorption with other gases such as CO_2 and has limited selectivity. From the point of view of engineering application, both of the adsorption selectivity and adsorption capacity of NO_x are very important for the adsorption and separation of atmospheric pollutant NO_x under complex atmosphere.

The MOFs materials have a developed pore structure, their skeleton can be chemically modified, and this is a higher design and adjustability in structure and performance ^[17-18]. As a catalyst carrier, the active center is introduced into MOFs through impregnation, deposition or adsorption to prepare highly dispersed metal or metal oxide nanoparticles with high stability, which greatly improves the adsorptive and catalytic performance of the active sites of metals^[19-21]. For example, Pd nanoparticle dispersed in the pores of Pd/MIL-101, and in the process of Heck reaction of acrylic acid and selective hydrogenation of styrene, nano-

Pd can hardly drop from MIL-101 with excellent catalytic reaction performance^[22]; Pd@Cu-BDC coupling reaction^[23], Pt@UIO-66-NH₂ dimensional selective hydrogenation ^[24], NiMo@MIL-101^[25] and Ru-PT@UIO-67 ^[26]photocatalytic hydrogen evolution, etc. The high dispersion and high stability of the metal active sites, which were customized by anchoring in the MOFs material, showed excellent catalytic performance. In view of this, it is of great application significance to further improve the adsorption and separation performance of MOFs to NO gas by improving the loading of metal nanoparticles on MOFs materials.

MOF-177, as a MOFs material with small density, large specific surface area, uniform pore distribution and good thermal stability, has shown unique advantages in gas adsorption and separation^[27-31]. According to the molecular simulation research results of Smit et al. ^[29], at 298 K and 0.1MPa, the NO adsorption capacity of MOF-177 can reached about 0.5mmol*g⁻¹, but the adsorption capacity of CO₂ was also high, reached about 1.0mmol*g⁻¹, and the adsorption selectivity of NO was low. Because the MOF-177 was weak in O₂ and N₂ adsorption and can serve as a good NO adsorption carrier, moreover Rh active component supported on oxide carrier showed excellent selective performance for the catalytic elimination of NO_x pollutants ^[32-33]. Therefore, if Rh is introduced into MOF-177 material to improve its NO adsorption selectivity, NO adsorption performance should be improved. This paper intends to combine noble metal Rh with porous MOF-177 material to form a composite material and explores its adsorption performance of NO in flue gas, so as to provide reference information for the purification and resource recovery of atmospheric pollutant NO.

1 Experimental Section

1.1Material preparation

(1)Carrier preparation

MOF-177 was synthesized by solvent-thermal method^[27]. 0.950 g (3.193 mmol) hexahydrate zinc nitrate $(Zn(NO_3)_2*6H_2O, AR)$, 0.190 g(0.433 mmol), 1, 3, 5-trisol (4-carboxyl phenyl) benzene (H₃BTB,AR), and 50 mL (99%) N, N-diethylformamide (DEF) were mixed into 100mL Teflon vessel and stirred until the solid was completely dissolved and the solution was pale yellow. The vessel was put into a vacuum drying oven at room temperature for 30 min before vacuum degassing, and then slowly put into a Teflon reaction kettle to prevent the influence of air on the reaction. The reaction kettle was placed in an electric thermostatic blast drying oven for 48 hours at 90 degC, and the temperature was cooled at 0.1 degC*min⁻¹rate. After taken out, the synthetic product was washed three times by ultrasonic oscillation with adding 10mL DEF, and then separated by centrifugation. The product was continually washed for 24 h with adding 20mL chloroform (CHCl₃) in a water bath at 70 degC to remove the residual impurities. After removing the chloroform, the product was dried in an oven to obtain a light yellow MOF-177 crystal \circ

(2)Noble metal element loading and sample reduction

Taking the above MOF-177 product as carrier, x Rh/MOF-177 (x = 1%, 2%, 3%, mass fraction) was prepared by impregnation method. Taking 3% Rh/MOF-177 as an example, 0.91 g 9% Rh(NO₃)₃ solution was diluted with distilled water in a beaker, then 0.97 g MOF-177 was added into the solution and mixed well at room temperature, the mixture was stirred for 4h, then centrifuged. The solid was dried and calcinated for 2 h under air atmosphere to get 3% Rh/MOF-177 powder.

0.1 g Rh/MOF-177 powder was put into the U-shaped quartz reaction tube and reduced for 2 h at 250°C under hydrogen flow. The reduced xRh/MOF-177 was denoted as R- x Rh/MOF-177 (x = 1%, 2%, 3%, mass fraction).

1.2NO pulse adsorption

Pulse adsorption method was used to measure the pure gas adsorption capacity of Rh/MOF-177, gas signal was detected by gas chromatographic TCD detector, and gas group components were calibrated and calculated from chromatographic peak by external reference method. With the maximum pulse gas peak (i.e. adsorption saturation) as the base, the reduction amount of each pulse peak was the single adsorption amount, and the accumulation of the adsorption amount was the pure gas adsorption capacity. The specific process was as follows:

0.1 g sample was placed to the U-shaped quartz reaction tube, and He carrier gas flow was purged at 200 °C until the chromatographic TCD detector baseline was stable. After the temperature dropped to room temperature (about 25 °C), the equal flow of adsorbed gas (NO, CO₂, O₂, N₂) entering the reaction tube with each pulse was controlled through a six-way valve until the peak area does not change. The numerical value of gas adsorption capacity for each pulse was calculated by the difference between the integral peak area of each pulse and the peak area at adsorption saturation. The peak area of gas was calibrated by trace injection needle. For example, when 1 μ L NO was injected, the peak area was 0.11, marked as A₀ = 0.11. The adsorption capacity was calculated as follows:Finally, the adsorption capacity of gas was

Gas Storage Capacity = $\frac{\mathbf{n}\mathbf{A}-\sum_{i=1}^{n}A_{i}}{A_{0}m}\times10^{-3}~\mathrm{mL}g^{\text{-1}}$

A was the peak area of pulse saturation, A_i was the peak area of ith pulse, A_0 was the peak area calibrated, n was the number of pulses, m was the quality of sample.

1.3 NO selective adsorption under mixed atmosphere

In simulated flue gas atmosphere (NO, CO_2 , O_2 , N_2 mixture), the adsorption capacity of NO selectivity of Rh/MOF-177 was tested by DECRA quantitative gas analysis mass spectrometer (QGA) from Hiden company of UK. The experimental method was as follows:

The empty U-shaped quartz tube was filled with N₂ flow at 200 °C to drive away other gases. When the temperature of the reaction tube dropped to room temperature, the gas flow was switched from the reaction tube to the bypass, and substituted for 20 ml·min⁻¹ gas mixture (1000 ppm NO, 1000 ppm O₂, 2000 ppm CO_2 , N₂remainder) though by-pass until QGA baseline smoothly. Then the gas mixture flow was switched to reaction tube with recording the signal of QGA. Similarly, 0.1g R-3% Rh/MOF-177 sample was put into the U-shaped quartz reaction tube and underwent the experimental process with the same method. Adsorption and breakthrough curves of the sample were obtained after the signal of the empty tube was deducted by the signal of QGA.

1.3Material characterization

The X-ray small-angle diffraction spectra of the samples were obtained using a PANalytical X'Pert³ powder diffractometer with a PIXcel detector. The X-ray generator was operated at 40 kV and 40 mA, using the CuK α ray at 1.54056 Å as the radiation source. Samples were scanned from 3° to 80° (2 ϑ) at scanning speed of 0.1347 °·s⁻¹with a stage size of 0.02626°. Divergence slit: Fixed slit 1/32°; incident beam path anti-scatter slit: Fixed slit 1/16°; diffraction beam path anti-scatter slit: AS slit 7.5mm. The instrument used for the XPS was ESCALAB 250Xi with AlK $_{\alpha}$ as the X source and C1s= 284.8EV as the standard pollution peak to calibrate the sample charge effect. Crystal structure and morphology of the sample were characterized by high-resolution transmission electron microscope (TEM), Jeol-2100F type, under 200kV accelerating voltage. The sample was pre-treated: dissolved in absolute ethyl alcohol and dispersed in the ultrasonic instrument for 15min. The fully dispersed sample liquid drops were dropped on the copper network and dried with an infrared lamp.

2 Results and Discussion

2.1 Structure and thermal stability of the synthesized products

MOF-177 was synthesized according to the method in reference^[27], and Rhodium component was carried on it. Figure 1 showed the XRD pattern of MOF-177, the main peaks at 5.7° and 10.5° were well identified. The XRD structure was consistent with that in reference ^[27], indicating that the samples prepared in this work should be MOF-177 frame structure compounds. However, crystal lattice changes slightly such as loaded crystal particles become smaller due to the peak at 10.5° was wider than without load samples in XRD pattern of 1%- 3% Rh/MOF-177. The weight loss of MOF-177, 3% Rh/MOF-177 and R-3% Rh/MOF-177 measured by the thermo-gravimetric method (TG) under N₂atmosphere were shown in Figure 2. It can be seen that the weight loss curves of the three MOFs were basically the same, and the thermal stability of MOF-177 was higher than that of 3% Rh/MOF and R-3% Rh/MOF-177. The weight loss process was divided into three stages, 3% Rh/MOF-177 and R-3% Rh/MOF-177 lost 10% and 6.6% respectively between 50 °C and 331 °C, and MOF-177 lose 15.8% between 50 °C and 427 °C which was 96 °C higher than the thermal stability of the material after loading. The first two stages of weight loss could be attributed to the volatilization of guest molecules H₂O and DEF. There were obvious differences among the three in the third stage. A drastic weight loss by heating from 427 °C ~540 °C was observed due to decomposition of MOF-177. 3% Rh/MOF-177 and R-3% Rh/MOF-177 gradually decomposed and collapsed at 331 °C ~450 °C. TG results showed that better thermal stability of MOF-177 and the loaded Rh/MOF-177 were all above 330 °C.

2.2Adsorption of NO by Rh/MOF-177

MOF-177 has been proved to have a relatively limited adsorption capacity by Smit ^[29] and Saha ^[30] et al. according to GCMC molecular simulation technology. Figure 3 showed adsorption capacity of NO on MOF-177 measured by pulse method at 298K. Through the peak area integral, it can be seen that MOF-177 had a trace adsorption of NO, which was due to its high specific surface area and large porosity ^[34-37]. NO adsorption capacity of MOF-177 material by calculating was $3.3 \times 10^{-2} \text{mL} \cdot \text{g}^{-1} (1.35 \times 10^{-3} \text{mmol} \cdot \text{g}^{-1})$. This was smaller than the results reported in literature molecular simulation (at 298 K and 0.1 MPa) for 0.5 mmol \cdot \text{g}^{-1} [^{29]}. It can be speculated that partial pressure of the adsorbent in He airflow decreased and was too smaller after the pulsing of NO.

Excellent catalytic performance was showed in removal of NO_x pollutants of Rh active component supported on oxide carriers ^[32-33]. In this work, Rh/MOF-177 was prepared with different Rh loadings and tested the adsorption performance of NO at room temperature in order to improve the NO adsorption and separation performance of MOF-177 in flue gas. The results were shown in Figure 4, Figure 5 and Table 2. It can be seen from Figure 4 and Figure 5 that the adsorption capacity of R-1% Rh/MOF-177 and R-3% Rh/MOF-177 was basically close to that of 1.24 ml·g⁻¹ and 1.18 ml·g⁻¹ respectively when the sample was injected in the first pulse. While the adsorption capacity of R-2% Rh/MOF-177 was 1.98 ml·g⁻¹, which was about 1.6 times that of R-1% Rh/MOF-177. During the 2-4 injection periods, the adsorption rate of R-1% Rh/MOF-177 gradually decreased, and the adsorption amount remained unchanged after the fifth injection, indicating that the adsorption was saturated, the total NO adsorption amount of R-1% Rh/MOF-177 was 2.57 mL·g⁻¹. The adsorption rate of R-2% Rh/MOF-177 was the largest among all the samples before the fourth pulse, but after the fourth pulse, it gradually decreased and lowered than R-3% Rh/MOF-177. After the seventh pulse, the curve was basically stable, R-3% Rh/MOF-177 reached its saturation state, the total adsorption volume of NO was 5.45 mL·g⁻¹. R-3% Rh/MOF-177 adsorbed NO at most, up to 7.55 mL·g⁻¹. It can be seen that the adsorption performance of NO on the loaded Rh of MOF-177 was greatly improved. In addition, the adsorption capacity of unreduced 3% Rh/MOF-177 to NO had only $0.12 \text{ mL} \cdot \text{g}^{-1}$ which was not significantly increased compared with MOF-177. The adsorption performance of R-Rh/MOF-177 after the reduction of H₂ at 250 °C for 2 h was significantly improved compared with that of non-reduced materials, which indicated that NO adsorption occurred on metallic Rh particles rather than Rh ions. A large amount of NO gas was adsorbed on R-3% Rh/MOF-177 whom has the strongest adsorption performance after 16 pulses in total. The result showed Rh/MOF-177 can improved the NO adsorption performance of the material. With the increasing of the Rh loading, the capacity of NO adsorption of R-Rh/MOF-177 was gradually enhanced as well as the performance of NO adsorption was significantly improved. From the table 2, it can be seen the adsorption capacity of NO on R-Rh/MOF-177 was basically proportional to the Rh loading capacity, and was much larger than samples without reduction as well as MOF-177 after converted the total NO adsorption volume unit for each sample into a molar value. So it was clear that there should be a directly relate between total adsorption value of NO on R-Rh/MOF-177 and the quantity of metallic Rh components loading.

XPS and TEM were used to characterize the samples in order to explore the structure-function relationship between the chemical and dispersion state and NO adsorption characteristics of R-Rh/MOF-177. Figure 6 was the XPS spectrum of Rh 3d state electron energy level. The peak separation of Rh 3d spectrum showed that the binding energy of Rh/MOF-177 was located at 304-320 eV, For Rh $3d_{5/2}$ spectrum, Rh³⁺ and Rh⁰ peaks occur at 309.0 eV and 307.8 eV, respectively^[36]. In Table 3, the relative contents of Rh components in different valence states were obtained by calculating the peak area. From Table 3 it can be seen that Rh³⁺ was the main existing form in the unreduced Rh/MOF-177 samples, and Rh⁰ appears after reduction. After reducing, the proportion of Rh⁰ was 29.4%[~] 36.2% and Rh³⁺ was 63.8[~] 70.6%. With the increasing of Rh loading, the content of Rh⁰ gradually increased. Combined with NO adsorption characteristics, it can clearly demonstrate that metallic Rh⁰ made a major role in NO adsorption after reducing by XPS results. The loaded Rh provided more unsaturated metal sites for gas adsorption of materials, then Rh⁰ played a major role in adsorption which reduced direct decomposition activation energy of NO greatly^[38]. The molar ratios of adsorption values for Rh⁰ and NO can be obtained by calculating (Table 3). The adsorption ratios of R-1% Rh/MOF-177, R-2% Rh/MOF-177 and R-3% Rh/MOF-177 were basically the same for 1:3, 1:4 and 1:3.3 respectively. Overall, three NO molecules were adsorbed by one reducing state of Rh⁰ approximately considering the experimental error.

Figure 7 a, b and c were TEM images of MOF-177 (under 20 nm scale), R-1% Rh/MOF-177(under 10 nm scale) and R-3% Rh/MOF-177(under 10 nm scale) respectively. It can be seen that the black dot shadow in Figure 7 b and c should be the Rh metal particle by comparing Figure 7 a. The comparison among Figure 7 a, b and c turned that the metal particle was homogeneous distribution and the diameter of Rh metal particle was approximately the same as the diameter of Rh atom, about 1.6 nm measured by DM (Digital Micrograph) software. With the increasing of the Rh, the density of shadow increased, but it still distributed evenly. It could indicate that the metal particles did not agglomerate after loading Rh, and the dispersion was relatively stable after the reduction at 250 °C. Form Figure 7 d-g element mapping, Rh element was marked as red and Zn was represented by purple. When the load only 1%, the Zn element was dense and Rh was dispersed; when the load was increased to 3%, the overall density of Zn element and Rh element was much higher than that of samples 1% loading. Because of the gap between the crystal particles of MOF-177, Rh element was dense and Zn element was dispersed at the graph. As there was no Zn element but Rh element in MOF-177. By observing TEM and element mapping, MOF-177 as a carrier can make the Rh components have homogeneous distribution and highly stable. In addition, the larger specific surface area and porosity provided a better condition for the adsorption of NO for R-Rh/MOF-177

2.3 NO adsorption selectivity

Although R-2% Rh/MOF-177 showed excellent NO adsorption capacity in the first three injections, in terms of the overall adsorption capacity, R-3% Rh/MOF-177 was superior. Through calculation, R-3% Rh/MOF-177 had the largest adsorption of NO, the total of 7.55 ml·g⁻¹. Therefore, R-3% Rh/MOF-177 was selected to explore the adsorption selectivity of NO at room temperature and atmospheric pressure. Figure 8 showed the pulse adsorption of O₂, CO₂ and N₂respectively by MOF-177 and R-3% Rh/MOF-177 at the same pressure and temperature. Figure 9 showed the pulse adsorption trend of MOF-177 (left) and the pulse adsorption trend of R-3% Rh/MOF-177 (right). Combined with the two figures, it can be seen that MOF-177 basically does not adsorbed N₂ and CO₂, and its adsorption capacity for NO was also relatively low. The adsorption of CO₂ and N₂ by R-3% Rh/MOF-177 after loading Rh showed little change, and the adsorption amount of MOF-177 was negligible compared with MOF-177. The adsorption amount of O₂ was increased, which was about 1.26 mL·g⁻¹ after calculation. However, compared with the 7.55 mL·g⁻¹ adsorption amount of NO by R-3% Rh/MOF-177, its O₂ adsorption was relatively weak. The single component pulsed adsorption results showed that R-Rh/MOF-177 had the best selective adsorption effect on NO.

The selective adsorption of NO by R-3% Rh/MOF-177 in NO, CO₂, O₂, N₂ mixture (1000 ppm NO, 1000 ppm O₂, 2000 ppm CO₂, N₂ remainder) was verified by DECRA quantitative gas analysis mass spectrometer. 0.1 g R-3% Rh/MOF-177 was put into the U-shaped reaction tube. After N₂ pretreatment, 20 mL·min⁻¹ mixture was injected. The adsorption curve and penetration curve were shown in Figure 10 (a) and Figure 10 (b), respectively. It can be seen from the figure, the signal changed in 8 min after the mixture was

fed into the reaction tube, O_2 and CO_2 were adsorbed and saturated in 12 min and 16 min respectively, and their concentrations quickly return to the intake value. The adsorption capacities of O_2 and CO_2 were 4.9×10^{-3} and 6.9×10^{-3} mmol·g⁻¹ respectively. During10-12 min, NO signal was at its lowest and appears on the platform, and it slowly returns to the intake gas at 12-45 min. The concentration was close to adsorption saturation, and the adsorption capacity reached about 100×10^{-3} mmol·g⁻¹, which was 15-20 times that of O_2 and CO_2 . Although adsorption experimental results of NO, CO_2 , O_2 , N_2 under the mixture had decreased, its adsorption characteristics and the pulse results of Rh/MOF-177 for NO, CO_2 , O_2 were consistent with single component adsorptive properties. A little of CO_2 and O_2 were absorbed, but large quantity of NO was collected. These results further verified the selectivity to the adsorption of NO. By calculation, the adsorption selectivity of NO in mixed atmosphere was about 89.45%. The experimental results showed that greatly enhancement of NO adsorption capacity and adsorption selectivity by introducing the Rh to MOF-177 material, and there were significant improvements of NO adsorption performance. As for NO_x in flue gas adsorption separation and then to stripping oxidation of absorbing compounds nitrate, this result should have large potential of air pollutants purification and recycling application of NO.

2.4 The structure-function relationship

The experimental results of this work showed that the adsorption property of NO was significantly improved after Rh was introduced into MOF-177, showing high NO adsorption capacity and adsorption selectivity. The pulse-method adsorption capacity of NO for the Rh/MOF-177 with a load of 3% Rh can reached about 310×10^{-3} mmol·g⁻¹, far higher than O₂ and CO₂ adsorption amount (about 15 to 20 times more), NO adsorption selectivity was up to near 90% under mixed atmosphere, this was just only on the 30% reduction degree of Rh components and the lower adsorption concentration or partial pressure conditions. Oxides such as adsorption materials supported on Al₂O₃ and zeolite carrier, its NO adsorption capacity under atmospheric pressure was usually 0.06~0.096 mmol·g^{-1[40]}, which was relatively limited. This indicated that MOF-177 material was an excellent carrier, which can provide better NO adsorption performance by introducing the active component Rh and so on, showed good carrier function. Because the thermal stability of Rh/MOF-177 was good to reach above 330 °C, this also created the application basis for the adsorption separation and enrichment of NO_x in the flue gas.

In general, the adsorption of NO by transition metal components loaded on oxide carriers is usually manifested as surface disproportionation reaction adsorption, and the formation of species such as N_2O_4 , NO_2 , NO_3 - or NO_2 - promotes the adsorption of NO ^[41-42], and the higher the valence state, the weaker the adsorption of NO ^[43]. Species nitroso Rh(NO) and dinitro Rh(NO)₂ are formed when NO is absorbed by Rh component with Al₂O₃ as the carrier ^[39-40,44]. The experimental results of this work showed that about one reduced state Rh⁰ on Rh/MOF-177 adsorbed three NO molecules, and it could be speculated that NO adsorption on Rh/MOF-177 formed an adsorptive state trinitro group (Rh(NO)₃) species, which was different from the results of NO adsorption machine in the existing literature. The outer-shell electron structure of Rh atom is $4d^85s^1$, and there are three unpaired electrons in the 4d and 5s orbitals, so it may be that the reduced state Rh⁰ on Rh/MOF-177 adsorbed NO and formed linear Rh-NO or Rh-N=O adsorption modes. The generation of these species resulted the higher adsorption capacity of Rh/MOF-177 to NO gas.

Some MOFs materials showed a better adsorption effect to pure NO molecule through opening various metal sites and amine group functionalization. In general, NO and unsaturated metal sites typically form 1:1 coordination, and there is also possible to form N-diazeniumdiolate(NONOate)structure through adsorptive chemical reactions between functionalized amine groups and NO, and then the adsorption was performed on the metal sites, exhibiting larger NO adsorption capacity, and there are NO chemical adsorption and physical adsorption on MOFs materials ^[7-9, 12-14]. However, when most MOFs materials adsorb NO, they will also absorb more CO₂, with poor selectivity^[15-16]. MOF-177 is composed of Zn ionic metal nodes and 1,3, 5-trisol (4-carboxyl phenyl) benzene (H₃BTB) ligand. The calculation results of molecular simulation also showed that MOF-177 had a certain adsorption capacity for NO under certain conditions, and its adsorption for CO₂ and O₂ was very limited^[29-30]. The experimental results of this work also verified this phenomenon. According to the results of high-resolution transmission electron microscope (TEM) and

Rh3delectron energy level XPS, the topological space structure of Rh/MOF-177 can be inferred as shown in Figure 11. With different from the reported NO adsorption mechanism of these MOFs materials, the NO adsorption model of Rh/MOF-177 materials changed greatly after the loading of Rh component—that is, one reduced Rh⁰ adsorbed three NO molecules and the NO adsorption form on Rh/MOF-177 should be a trinitrogroup (Rh(NO)₃) adsorbed species. This resulted in significantly enhanced NO adsorption performance of Rh/MOF-177, which not only NO adsorption capacity significantly increased but also selectivity increased. According to the results of high-resolution transmission electron microscope (TEM), the Rh particles reduced at 250 degC with different load of Rh components were basically atomic-sized dispersion state with good stability. This was due to the larger specific surface area and porosity of MOF-177, which provided a better carrier function for the high dispersion and stability of Rh component.

3 Conclusion

The experimental adsorption capacity of MOF-177 for NO, O_2 and CO_2 by pulse method was very small and N_2 was hardly adsorbed at room temperature, which was consistent with the molecular simulation calculation results in the literature. The introduction of Rh component in MOF-177 can significantly improve its NO adsorption performance, showed high NO adsorption capacity and adsorption selectivity. The NO adsorption capacity increased with the increase of Rh load and was related to the reduced state Rh⁰. The adsorption capacity of NO by pulse method can reach about $310 \times 10^{-3} \text{ mmol}^{\circ} \text{g}^{-1}$ when 3% Rh/MOF-177 was reduced for 2 h at 250 degC. The adsorption capacity of NO in simulated flue gas atmosphere was about 15-20 times higher than that of O_2 and CO_2 , and the adsorption selectivity was up to near 90%.

The experimental results showed that one reduced state Rh^0 adsorbed three NO molecules, and the NO adsorption on Rh/MOF-177 should be the adsorption state trinitro-group ($Rh(NO)_3$) species. After reduction at 250 degC, the Rh particles in the MOF-177 exhibited atomic-size dispersion with good stability, and the MOF-177 material showed good carrier function.

Reference

[1] Busca G, Lietti L, Ramis G, Berti F. Chemical and mechanistic aspects of the selective catalytic reduction of NO_x by ammonia over oxide catalysts [J]. Applied Catalysis B: Environmental. 1998, 18(1-2): 1-36. [2] Chen M-Y, Zhao M-M, Tang F-S, Ruan L, Yang H-B, Li N. Effect of Ce doping into V₂O₅-WO₃/TiO₂catalysts on the selective catalytic reduction of NO_x by NH₃. Journal of Rare Earths. 2017, 35(12): 1206-1215. [3] Li J-N, Han X, Zhang X, Sheveleva AM, Cheng YQ, Tuna F, McInnes Eric JL, Laura JM, Teat SJ, Luke LD, Ramirez AJ., Schroder M and Yang S-H. Capture of nitrogen dioxide and conversion to nitric acid in a porous metal–organic framework. Nature Chemistry. 2019, 11: 1085–1090. [4] Tang F-S, Zhuang K, Yang F, Yang L, Xu B-L, Qiu J-H, Fan YN. Effect of dispersion state and surface properties of supported vanadia on the activity of V₂O₅/TiO₂ catalysts for the selective catalytic reduction of NO by NH₃. Chinese Journal of Catalysis. 2012, 33(6): 933-940.

[5] Zhang Z-Q, Atkinson JD, Jiang B-Q, Roodb MJ, Yan Z-F. Nitric oxide oxidation catalyzed by microporous activated carbon fiber cloth: An updated reaction mechanism [J]. Applied Catalysis B: Environmental. 2014, 148(27): 573-581.

[6] Xiao B, Wheatley PS, Zhao X-B, Fletcher AJ, Fox S, Rossi AG, Megson I L, Bordiga S, Thomas L, and Morris RE. High-capacity hydrogen and nitric oxide adsorption and storage in a metal-organic framework. [J]. Journal of the American Chemical Society. 2007, 129(5): 1203-1209.

[7] McKinlay AC, Xiao B, Wragg D, Wheatley PS, Megson IL., and Morris RE. Exceptional behavior over the whole adsorption-storage-delivery cycle for NO in porous metal organic frameworks [J]. Journal of the American Chemical Society. 2008, 130(31): 10440-10444.

[8] Cattaneo D, Warrender SJ, Duncan MJ, Castledine R, Parkinson N, Haley I and Morris RE. Water based scale-up of CPO-27 synthesis for nitric oxide delivery [J]. Dalton transactions. 2016, 45(2): 618-629.

[9] Barth B, Mendt M, Poppl A, Hartmann M. Adsorption of nitric oxide in metal-organic frameworks:

Low temperature IR and EPR spectroscopic evaluation of the role of open metal sites [J]. Microporous and Mesoporous Materials. 2015, 216: 97-110.

[10] Bonino F, Chavan S, Vitillo JG, Groppo E, Agostini G, Lamberti C, Dietzel PD, Prestipino C, and Bordiga S. Local structure of CPO-27-Ni metal organic framework upon dehydration and coordination of NO [J]. Chemistry of Materials: A Publication of the American Chemistry Society. 2008, 20(15): 4957-4968.

[11] Li B, Wen H-M, Cui Y-J, Zhou W, Qian G-D, Chen B-L. Emerging multifunctional metal-organic framework materials. Advanced Materials. 2016, 28: 8819–8860.

[12] Dietzel PD, Morita Y, Blom R, and Fjellvag H. An in situ high-temperature single-crystal investigation of a dehydrated metal–organic framework compound and field-induced magnetization of one-dimensional metal–oxygen chains [J]. Angewandte Chemie. 2005, 44(39): 6354-6358.

[13] Dietzel PD, Panella B, Hirscher M, Blom R and Fjellvag H. Hydrogen adsorption in a nickel based coordination polymer with open metal sites in the cylindrical cavities of the desolvated framework [J]. Chemical Communications. 2006, 9: 959-961.

[14] Nguyen JG, Tanabe KK and Cohen SM. Postsynthetic diazeniumdiolate formation and NO release from MOFs [J]. CrystEngComm. 2010, 12(8): 2335-2338.

[15] Meng G-H, SongX-D, Ji M, Hao J-Y, Shi Y-T, Ren S-Z, Qiu J-S, Hao C. Molecular simulation of adsorption of NO and CO₂mixtures by a Cu-BTC metal organic framework [J]. Current Applied Physics the Official Journal of the Korean Physical Society. 2015, 15(9): 1070-1074.

[16] Tan K, Zuluaga S, Gong Q, Gao Y-Z, Nijem N, Ji L, Thonhauser T, and Chabal YJ. Competitive coadsorption of CO₂ with H₂O, NH₃, SO₂, NO, NO₂, N₂, O₂, and CH₄ in M-MOF-74 (M = Mg, Co, Ni): the role of hydrogen bonding. Chemistry of Materials. 2015, 27(6): 2203-2217.

[17] Li B, Wen H-M, Yang Y Cui Y-J, Zhou W, Chen B-L, Qian G-D. Nanospace within metal-organic frameworks for gas storage and separation. Materials Today Nano. 2018, 2: 21-49.

[18] Pei J-Y, Shao K, Wang J-X, Wen H-M, Yang Y, Cui Y-J, Krishna R, Li B and Qian G-D. A chemically stable hofmann-type metal-organic framework with sandwich-like binding sites for benchmark acetylene capture. [J]. Advanced Materials. 2020, 32: 1908275.

[19] Tang F-S, Zhao J and Chen B-L. Porous coordination polymers for heterogeneous catalysis. Current Organic Chemistry. 2018, 22(18): 1773-1791.

[20] Huang H-G, Shen K, Chen F and Li Y-W. Metal-organic frameworks as a good platform for the fabrication of single-atom catalysts[J]. ACS Catalysis. 2020, 10: 6579-6586.

[21] Chen X-D, ShenK, Ding D-N, Chen J-Y, Fan T, WuR-F, Li Y-W. Solvent-driven selectivity control to either anilines or dicyclohexylamines in hydrogenation of nitroarenes over a bifunctional Pd/MIL-101 catalyst[J]. ACS Catalysis, 2018, 8(11): 10641-10648.

[22] Henschel A, Gedrich K, Kraehnert R and Kaskel S.Catalytic properties of MIL-101.Chem. Comm. 2008, 44(35): 4192-4192. [23] Annapurna M, Parsharamulu T, Reddy VP, Suresh M, Likhar PR and Kantam ML. Nano palladium supported on high-surface-area metal-organic framework MIL-101: an efficient catalyst for Sonogashira coupling of aryl and heteroaryl bromides with alkynes. Applied Organometallic Chemistry. 2015, 29(4): 234-239. [24] Chen Y-F, Tan L-L, Liu J-M, Qin S, Xie Z-Q, Huang J-F, Xu Y-W, Xiao L-M, Su C-Y. Calix[4]arene based dye-sensitized Pt@UiO-66-NH₂ metal-organic framework for efficient visible-light photocatalytic hydrogen production. Applied Catalysis B: Environmental. 2017, 206, 426–433. [25] Zhen W-L, Gao H-B, Tian B, Ma and Lu G-X. Fabrication of low adsorption energy Ni-Mo cluster cocatalyst in metal-organic frameworks for visible photocatalytic hydrogen evolution. ACS Appl. Mater. Interfaces. 2016, 8(17): 10808-10819. [26] Hou C-C, L T-T, Cao S, Chen Y, and Fu W-F. Incorporation of a [Ru(dcbpy)(bpy)₂]²⁺photosensitizer and a Pt(dcbpy)Cl₂ catalyst into metal-organic frameworks for pho-

to catalytic hydrogen evolution from aqueous solution [J]. Journal of Materials Chemistry A. 2015, 3 (19): 10386–10394.

[27] Chae HK, Siberio-Pe'rez DY, Kim J, Go YB, Eddaoudi M, Matzger A J, O'Keeffe M, Yaghi OM. A route to high surface area, porosity and inclusion of large molecules in crystals [J]. Nature. 2004, 427(6974): 523-527.

[28] Li Y-W, Yang RT. Gas adsorption and storage in metal-organic framework MOF-177. [J] Langmuir. 2007, 23(26):12937-12944.

[29] Sun W-Z, Lin L-C, Peng X, Smit B. Computational screening of porous metal-organic frameworks and zeolites for the removal of SO₂ and NOx from flue gases. [J]. AIChE Journal. 2014, 60(6): 2314-2323.

[30] Saha D, Bao Z, Jia F and Deng S-G. Adsorption of CO₂, CH₄, N₂O, and N₂ on MOF-5, MOF-177, and zeolite 5A. Environmental Science and Technology. 2010, 44(5): 1820-1826.

[31] Yang K, Xue F, Sun Q, Yue R-L, Lin D-H. Adsorption of volatile organic compounds by metal-organic frameworks MOF-177 [J]. Journal of Environmental Chemical Engineering. 2013, 1(4): 713-718.

[32] Cao Y-D, Ran R, Wu X-D, Zhao B-H and Weng D. Improved activity and durability of Rh-based threeway catalyst under diverse aging atmospheres by ZrO₂ support [J]. Journal of Environmental Sciences. 2017, 52: 197-203.

[33] Fu M-L, Ye D-Q, Liang H. Effect of the Rhodium content on the performance of supported nobel metal three-way catalysts[J]. Preious Metals, 2002, 23(2): 6-10.

[34] Panella BB, Hirscher M, Putter H, and Muller U. Hydrogen adsorption in metal-organic frameworks: Cu-MOFs and Zn-MOFs compared [J]. Advanced Functional Materials. 2006, 16(4): 520–524.

[35] Furukawa H, Miller MA and Yaghi OM. Independent verification of the saturation hydrogen uptake in MOF-177 and establishment of a benchmark for hydrogen adsorption in metal-organic frameworks[J]. Journal of Materials Chemistry. 2007, 17(30): 3197–3204.

[36] Kondarides DI, Verykios XE. Effect of chlorine on the chemisorptive properties of Rh/CeO₂ catalysts studied by XPS and temperature programmed desorption techniques[J]. Journal of Catalysis, 1998, 174(1):52-64.

[37] Rowsell JLC, Yaghi OM. Metal-organic frameworks: a new class of porous materials [J]. Microporous and Mesoporous Materials. 2004, 73(1-2): 3-14.

[38] Tu X-Y, Tian K, Dai S-S. Density function study of NO adsorption and direct decomposition on Rh(100) and Rh(111) surfaces [J]. Chemical journal of Chinese universities. 2005, 26(12): 2354-2356.

[39] Newton MA, Dent AJ, Fiddy SG, Jyoti B and Evans J. Identification of the surface species responsible for N_2O formation from the chemisorption of NO on Rh/alumina [J]. Physical chemistry chemical physics, 2007, 9(2):246-249.

[40] Liang J, Wang H-P, Spicer L-D. FT-IR study of nitric oxide chemisorbed on Rh/Al₂O₃ [J]. The Journal of Physical Chemistry. 1985, 89(26): 5840-5845.

[41] Mul G, Perez-Ramirez J, Kapteijn F and Moulijin JA. NO Adsorption on ex-framework [Fe,X]MFI catalysts: novel IR bands and evaluation of assignments [J]. Catalysis Letters. 2002, 80(3): 129-138.

[42] Kang H-X, Guo Q-H, Li J-W, Li Y-X. The investigation of both NO and N₂O adsorption and NOassisted action to N₂O decomposition mechanism over Fe-ZSM-5 zeolite [J]. Journal of Henan University (Natural Science). 2008, 38(03): 261-265.

[43] Zhu C-Y, Fu J, Du L-C, Li Q-Z. Studies on the adsorption of NO on Fe(III)/Y zeolite by FT-IR method [J]. Chinese Science Bulletin. 1994, 39(22): 2055-2058.

[44] Li G-X, Huang Q-Z, Hou J. NO adsorption activity of Pt supported on activated carbon fiber [J]. Chinese Journal of Catalysis. 2003, 24(2): 107-110.

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